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Machine learning-based prediction of elastic properties of amorphous metal alloys

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ABSTRACT

The Young's modulus *E* is the key mechanical property that determines the resistance of solids to tension/compression. In the present work, the correlation of the quantity Ewith such characteristics as the total molar mass M of alloy components, the number of components *n* forming an alloy, the yield stress $\sigma_{\rm v}$ and the glass transition temperature T_{σ} has been studied in detail based on a large set of empirical data for the Young's modulus of different amorphous metal alloys. It has been established that the values of the Young's modulus of metal alloys under normal conditions correlate with such a mechanical characteristic as the yield stress as well as with the glass transition temperature. As found, the specificity of the "chemical formula" of alloy, which is determined by molar mass M and number of components n, does not affect on elasticity of the material. The machine learning algorithm identified both the quantities M and n as insignificant factors in determining E. A simple non-linear regression model is obtained that relates the Young's modulus with T_g and σ_v , and this model correctly reproduces the experimental data for metal alloys of different types. This obtained regression model generalizes the previously presented empirical relation $E \simeq 49.8\sigma_v$ for amorphous metal alloys.

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1. Introduction

Knowledge of the Young's modulus *E* plays a crucial role in the development of functional materials based on amorphous metal alloys and in ensuring the mechanical stability of structures in various industries [1–3]. The modulus *E* characterizes the elasticity of a solid material and its resistance to external mechanical influences. From a microscopic point of view, elasticity is defined by interatomic (intermolecular) interaction forces as well as structure of a solid [4–6]. In this regard, the elastic properties of the material must be determined by the type of chemical elements that form this material as well as by the ratio of the concentrations of these elements, i.e., in what proportions on relation to each other the different elements are presented in the material [7,8]. However, there are no simple relationships connecting these quantities to each other. The solution of this problem within the framework of the microscopic theory of elasticity is nontrivial.

In the case of amorphous solids, many of their physical (mainly, mechanical) properties are determined by the so-called glass forming ability (GFA) [9–13]. In other words, any amorphous solid is characterized by some GFA, which determines the ability of the corresponding melt to form an amorphous phase. Although there is no generally accepted GFA criterion,

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practically all known GFA criteria are determined through the glass transition temperature T_g [14–16]. In this regard, it seems quite reasonable to expect that the glass transition temperature T_g can be related to the elastic properties of amorphous solids. Thus, in the case of amorphous solids, in particular, amorphous metal alloys, we formulate the following question: is there a correspondence between the elastic modulus and some physical and chemical characteristics of solids, such as the yield stress σ_y , the molar mass M, the number of components n forming the material, and the glass transition temperature T_g ?

Young's modulus can be determined from the well-known "stress-strain" relation, which indicates correspondence between strain and stress for a material [17,18]. Here, the magnitude of *E* is determined from the slope of the initial linear part of the "stress-strain" relation corresponding to the elastic region [19,20]. Sometimes, it becomes necessary to estimate *E* from known physical characteristics without mechanical tests and computer simulations. This task is especially relevant in the case of fragile materials with a complex sample preparation procedure, in the case of synthesizing new materials with required mechanical properties, and in the case of predicting strength characteristics of existing materials under various thermodynamic conditions. It is noteworthy that modern artificial intelligence methods, in particular, machine learning, can be used to solve this task [21–25].

Machine learning methods have been repeatedly applied to predict the value of the Young's modulus of various materials including metal alloys [26], silicate glasses [27], polymeric materials [28] and rocks [29]. These studies have shown the efficiency of machine learning based on the artificial neural networks (ANN's) in estimating the Young's modulus from known physical characteristics of materials. For example, in Ref. [26], it was shown that the average concentration of valence electrons, the difference in atomic radii and the melting temperature are the key characteristics that determine the Young's modulus of refractory alloys. Despite the successes achieved, similar studies have not yet been carried out with respect to such an extensive class of materials as amorphous metal alloys.

In the present study, the calculation of the Young's modulus of amorphous metal alloys with different compositions and mechanical properties is performed by machine learning based on the ANN's. This calculation is carried out by set of experimental data for metal alloys based on Al, Au, Ca, Co, Cu, Fe, La, Hf, Mg, Ni, Pd, Pt, Re, Sr, Ti, W, Zr and rare earth elements [30]. As an input parameters in the ANN, we chose such physical characteristics as the total molar mass *M* of the alloy components, the number of components *n*, the yield stress σ_y , and the glass transition temperature T_g . Information about these characteristics is available in the case of the considered amorphous metal alloys. It has been established that these characteristics are sufficient to determine the Young's modulus with high accuracy.

2. Neural network construction and learning

In the present work, the ANN of direct propagation was built to perform the required calculations. This ANN consists of four layers (see Fig. 1). The first layer contains 5 input neurons. The next two layers are hidden and each of them contains ten neurons. The values of the Young's modulus *E* are determined by the one output neuron. The input neurons are supplied by values of the molar mass *M*, the number of components *n*, the yield stress σ_y , the glass transition temperature T_g and the noise factor for different metal alloys (see Tables 1–5 in Supporting Materials). All the necessary data for these alloys are taken from Ref. [30]. The values of the input physical quantities are preliminarily calibrated so that they change in the range from 0 to 1. For learning and testing of the ANN, we have used metal alloys, whose the Young's modulus *E* is known. For the remaining alloys, the values of *E* are predicted by the ANN after it has been learned.

The main working regime of the ANN is the direct propagation of information, in which the output value of each neuron is calculated as follows [31]

$$n_i^{(k)} = f(w_{11}^{(k-1)} n_1^{(k-1)} + w_{12}^{(k-1)} n_2^{(k-1)} + \dots + w_{ij}^{(k-1)} n_j^{(k-1)} + b_i^{(k)})$$
(1)

or

$$n_i^{(k)} = f\left(W_i^{(k)} + b_i^{(k)}\right),$$
(2)

where the input value of the *k*th layer neuron is calculated through the expression

$$W_i^{(k)} = \sum_{j=1}^{N_{k-1}} w_{ij}^{(k-1)} n_j^{(k-1)}.$$
(3)

Here, $n_i^{(k)}$ is value of the *i*th neuron in the *k*th layer (k = 1, 2, 3, 4); $w_{ij}^{(k-1)}$ is value of the weight of the (k - 1)th layer going from the neuron with index *j* to the neuron with index *i* from the *k*th layer; $b_i^{(k)}$ is the bias weight acting on *i*th neuron from the *k*th layer; N_{k-1} is the number of neurons in the (k - 1)th layer; f(...) is the sigmoid activation function

$$f(x) = \frac{1}{1 - e^{-x}},\tag{4}$$

which takes values from 0 to 1. The weights $w_{ij}^{(k-1)}$ and $b_i^{(k)}$ are assigned the fixed value 0.5 before the first iteration of the ANN.



Fig. 1. Scheme of the artificial neural network designed to calculate the Young's modulus *E* of metal alloys from the known values of the molar mass *M*, the number of components *n*, the yield stress σ_y and the glass transition temperature T_g , and also taking into account the noise factor (Noise).

For learning and minimizing the loss function of the ANN, the backpropagation and the gradient descent methods are applied [32]. These methods allow us to adjust the values of all weights taking into account the error between the output neuron and a required result (details are discussed in Supporting Materials in section "1. Artificial neural network learning method"). The criterion to finish learning is the passage of a certain number of epochs. The optimal number of epochs is determined by calculation of the mean square error (MSE), which in turn is a loss function for the considered ANN:

$$MSE = \frac{1}{N} \sum_{l=1}^{N} \left(n_1^{(4)}(l) - L_l \right)^2.$$
(5)

Here, N = 20 is the number of alloys in the test data set formed from different types of alloys. Fig. 2 shows that the result of Eq. (5) is the decaying curve, which start to take the minimal value MSE \simeq 0.0015 at after 2200 epochs. This number of epochs can be considered as optimal, since with a further increase in the number of epochs the MSE increases due to retraining of the ANN.

3. Results

After learning the ANN, test calculations were carried out and the most significant physical parameters were identified, the values of which with the values of the Young's modulus. For this purpose, the MSE was calculated for various combinations of physical parameters at the ANN input. As can be seen from Fig. 3, the smallest MSE is achieved when all four input physical parameters M, n, σ_y and T_g are taken into account and is equal to ~ 0.00178 . Addition of the noise factor leads to a slight increase in the error. Exclusion from consideration of the molar mass M and the number of components n also has no significant effect on the result and it increases the MSE up to $\simeq 0.00198$. Thus, it can be seen from Fig. 3 that the yield stress σ_y and the glass transition temperature T_g are the most significant parameters. This is also confirmed by the fact that taking only σ_y as an input factor leads to MSE $\simeq 0.00202$, while only with T_g the MSE is $\simeq 0.0137$. Further, M and n are identified as insignificant factors, the MSE of which is comparable with the error of the noise factor, ~ 0.058 . This result shows that the molar mass M and the number of components n do not reveal a correlation with the resistance of a material to mechanical deformation.



Fig. 2. Neural network loss function that is calculated using Eq. (5) according to test data set for 20 different metal alloys.



Fig. 3. MSE values at various combinations of the ANN input parameters.

For the considered alloys, there is a clear relationship between the yield stress σ_y and the Young's modulus *E*, that is confirmed by the results presented in Fig. 4(a). As seen, there is a linear relationship between *E* and σ_y , which can be reproduced by the function $E = 49.8\sigma_y$. This result confirms the known relation $E/\sigma_y \simeq 49.8$ obtained earlier in Ref. [30] on the basis of empirical data. We have found that the ANN result fits this linear relation even for alloys whose Young's modulus was not previously known [red dots in Fig. 4(a)]. This confirms the applicability of the ANN to predict the Young's modulus of various types of metal alloys based on the known values of the yield stress σ_y .

Fig. 4(b) shows that there is no simple relationship between *E* and T_g . The correspondence between these physical parameters is not reproduced by any simple law due to the fact that for a type of alloy there is a certain range of *E* and T_g values. Despite this, the prediction of the ANN fits into the resulting ranges of *E* and T_g [see Fig. 4(b)], which confirms the presence of a hidden relationship between the Young's modulus and the glass transition temperature.

To quantitative characterization of the hidden relationship between *E* and T_g as well as to obtain a general equation relating the quantities *E*, σ_y , T_g , we have studied the reproducibility of data in the framework of a non-linear regression model of the following form [22,33]:

$$Y(X_1, X_2) = a_0 + a_1 X_1 + a_2 X_2 + a_3 X_1 X_2 + a_4 X_1^2 + a_5 X_2^2 + a_6 (X_1 X_2)^2.$$
(6)



Fig. 4. (a) Correlation between the Young's modulus *E* and the yield stress σ_y , which obeys the linear law $E/\sigma_y \simeq 49.8$. (b) Correspondence between the Young's modulus *E* and the glass transition temperature T_g .

Here, $Y \equiv E$ is the output factor, while $X_1 \equiv \sigma_y$ and $X_2 \equiv T_g$ are the input factors. The values of the coefficients a_0, a_1, \ldots, a_6 were selected by enumeration from the range [-2000; 2000] with the increments ds = 500. For each combination of the values of these coefficients, the error value was calculated as follows

$$\xi = \sqrt{\frac{1}{N_{Exp}} \sum_{i=1}^{N_{Exp}} (Y_i - E_i)^2},$$
(7)

where Y_i is the Young's modulus calculated from relation (6), where the input factors σ_y and T_g take the experimental values. The quantity E_i characterizes the experimental values of the Young's modulus. $N_{Exp} = 173$ is the number of alloys for which the experimental values of σ_y , T_g , and E are known (see Tables 1–5 in Supporting Materials). The values of the coefficients $a_0^{(best)}$, $a_1^{(best)}$, \ldots , $a_6^{(best)}$ giving the best fit between the experimental Young's modulus and the result of Eq. (6) are determined at a minimal value of the error ξ . Further, the ranges of coefficient values are narrowed and set as $[a_0^{(best)} - ds; a_0^{(best)} + ds]$, $[a_1^{(best)} - ds; a_1^{(best)} + ds]$, \ldots , $[a_6^{(best)} - ds; a_6^{(best)} + ds]$. The increment ds is also halved. The search for the optimal values of the coefficients is repeated anew until the step size becomes ds < 0.001. The values of the fitting coefficients were determined as $a_0 = 25$ GPa, $a_1 = 41.4$, $a_2 = -0.0046$ GPa /K, $a_3 = 0.0015$ K⁻¹, $a_4 = a_5 = a_6 = 0$. Thus, on the basis of Eq. (6), we obtain equation, which relates the quantities σ_y , T_g and E:

$$E(\sigma_{\rm v}, T_{\rm g}) = 25 + 41.4\sigma_{\rm v} - 0.0046T_{\rm g} + 0.0015\sigma_{\rm v}T_{\rm g}.$$
(8)

In geometry, equation of this kind is known as the equation of a planar surface with a certain tilt.

Fig. 5 shows that the experimental and predicted values of the Young's modulus are indeed located on the planar surface either in its immediate vicinity; this surface is reproduced by Eq. (8). At the same time, the data predicted on the basis of machine learning are located closer to the planar surface shown in Fig. 5 compared to the experimental data. The experimental data are also located along this surface and follow its slope, whereas these data have a much larger scatter compared to machine learning predictions. For example, a significant divergence between the experimental data and the result of Eq. (8) is observed in the case of alloys based on tungsten and cobalt containing up to 32% of boron. Here, it should be noted that the glass transition temperature T_g included in Eq. (8) is not fixed for a specific alloy and, as a rule, depends on the cooling rate of the liquid melt. The yield stress σ_y also depends on the mechanical test conditions such as the strain rate as well as the direction and type of the applied force. Therefore, the absence of unified preparation and deformation protocols for various types of amorphous alloys may be one of the reasons for the divergence between the experimental data and the result of regression analysis.

4. Conclusions

Thus, based on a large set of empirical data for amorphous metal alloys of various compositions, a relationship has been established between the Young's modulus *E* and such key physical characteristics of alloys as the molar mass *M*, the number of components *n*, the yield stress σ_y , the glass transition temperature T_g . Using the machine learning based on the ANN's, it was found that the values of the Young's modulus of metal alloys under normal conditions correlate with σ_y and



Fig. 5. (a) Correspondence between the Young's modulus *E*, the yield stress σ_y and the glass transition temperature T_g . The experimental values of *E* (Exp) are compared with the Young's modulus predicted based on machine learning (ML). The result of a non-linear regression model (NRM) obtained by Eq (8) is presented as a planar surface. (b) This correspondence from a different angle showing that all of the data are located in the vicinity of the planar surface.

 T_g . As found, a non-linear regression model correctly reproduces the experimental data *E*, T_g and σ_y for different metal alloys, and this model generalizes the previously known empirical relation $E/\sigma_y \simeq 49.8$ for amorphous metal alloys [30]. At the same time, the results reveal that the molar mass *M* and the number of components *n* are not significant factors and do not affect on values of the Young's modulus.

The obtained results are of great practical importance, since they confirm the possibility of estimating the mechanical properties of metal alloys based on information about other physical properties, which can be used in the synthesis of alloys with a required properties. Moreover, the combination of the ANN and regression analysis can be applied to solve other similar tasks, where it is required to predict the physical parameters of various materials based on a limited set of training data [34]. For example, such tasks include determining the GFA of various types of materials [35,36], as well as the task of predicting the low-temperature characteristics of glass-formers (glass transition temperature, fragility index, etc.) from high-temperature data (viscosity, melting temperature, Arrhenius transition temperature, etc.) [25,37–39].

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

The data required to reproduce the results of the present study are available to download from Github: https://github.com/BulatGalimzyanov/MLdata.git.

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Appendix A. Supplementary data

Supplementary material related to this article can be found online at https://doi.org/10.1016/j.physa.2023.128678.

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