## Electron Paramagnetic Resonance of Gd<sup>3+</sup> Ions in Powders of LaF<sub>3</sub>:Gd<sup>3+</sup> Nanocrystals

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The observation of electron paramagnetic resonance of  $Gd^{3+}$  ions in nanosized powders of rare-earth fluorides  $LaF_3$ : $Gd^{3+}$  has been reported. The measurements have been performed on a single crystal and microand nanosized powders at room temperature. Electron paramagnetic resonance spectra and spin-Hamiltonian parameters of  $Gd^{3+}$  ions have been obtained. A qualitative difference of spectra in nano- and micropowders due to the increase in the spread of the crystal field parameters with the decrease in the particle size has been found. The relationship between the single-crystal domain size and the hydrothermal treatment time has been established.

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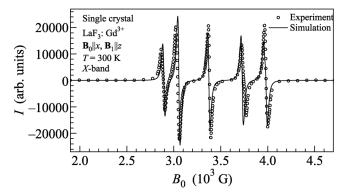
In this work, we studied nanosized powders of rareearth trifluoride  $LaF_3$  activated by  $Gd^{3+}$  ions. It is known that  $Gd^{3+}$  ions have no orbital angular momentum in the states of the lower  ${}^3S_{7/2}$  multiplet. Consequently, the spin—lattice relaxation time of these states is rather long for the electron paramagnetic resonance signal to be observed at room temperature. This is the origin of gadolinium compounds being promising contrast agents in magnetic resonance tomography [1-3]. At present, contrast agents on the basis of rareearth ions with the optimal structural and physicochemical properties making it possible to achieve the maximal contrast of the magnetic resonance tomography image are actively being sought [2, 4].

One of the simplest methods of the synthesis of nanosized powders is the method of deposition from colloid solutions. The structure and sizes of the synthe sized particles can be controlled by the variation of the temperature and time of hydrothermal treatment [5]. To this end, a colloid solution with nanoparticles is placed in a microwave oven (the radiation power of ~650 W, the treatment time  $\tau$  varying from 20 min to several hours). Recently, a new technique of varying the nanoparticle sizes of isostructural compounds (praseodymium trifluoride PrF<sub>3</sub>) by varying the hydrothermal treatment time was tested [6, 7]. The increase in the nanoparticle sizes and ordering of their crystal structure were observed with increasing τ. Note two more interesting effects directly associated with the specificity of the structure of PrF<sub>3</sub> nanoparticles. Water clusters were detected in PrF<sub>3</sub> nanoparticles using electron microscopy and nuclear magnetic resonance [8]. The correlation of the nuclear magnetic relaxation parameters of <sup>3</sup>He and nanoparticle sizes was revealed for PrF<sub>3</sub> nanopowders in contact with liquid helium-3. [9].

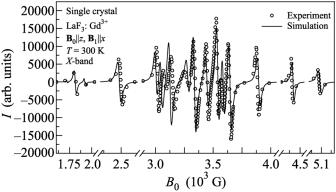
In this work, we present the spectral study of a series of LaF<sub>3</sub> samples with the content of 0.5% of impurity Gd<sup>3+</sup> ions. A single crystal was grown by the Bridgman-Stockbarger method. The micropowder was prepared by cutting a single crystal in a sapphire mortar and the subsequent sieving through a sieve. The particle size was 10-45 µm. Nanopowders were synthesized by the method of deposition from colloid solutions and subjected to the hydrothermal treatment. Lanthanum trifluoride was doped with gadolinium as follows. Gadolinium oxide (0.5%) was added to lanthanum oxide dissolved in nitric acid (99%). The prepared solution was carefully stirred. Then, the synthesis was performed as above [5]. Four samples nos. 1–4 were synthesized with treatment times  $\tau = 0$ , 20, 40, and 420 min, respectively. The X-ray analysis identified the samples as LaF<sub>3</sub>. Electron paramagnetic resonance spectra were measured on an ESP-300 spectrometer (Bruker) in the X band (9.6 GHz) at a temperature of 300 K. Figures 1–5 show the measurement results.

To interpret the measured spectra, we write the spin Hamiltonian of the ground state  ${}^8S_{7/2}$  of the Gd<sup>3+</sup> ion in the magnetic field  $\mathbf{B} = \mathbf{B}_0 + \mathbf{B}_1 \cos(\omega t)$ :

$$H = g\mu_{\mathrm{B}}\mathbf{B} \cdot \mathbf{S} + \sum_{\substack{p = 2, 4, 6, \\ k = -p, \dots, p}} B_{p}^{k} O_{p}^{k}, \tag{1}$$



**Fig. 1.** Electron paramagnetic resonance spectrum of  $Gd^{3+}$  ions in the LaF<sub>3</sub>: $Gd^{3+}$  single crystal ( $\mathbf{B}_0 \parallel x$  and  $\mathbf{B}_1 \parallel z$ ).



**Fig. 2.** Electron paramagnetic resonance spectrum of  $Gd^{3+}$  ions in the LaF<sub>3</sub>: $Gd^{3+}$  single crystal ( $\mathbf{B}_0 \parallel z$  and  $\mathbf{B}_1 \parallel x$ ).

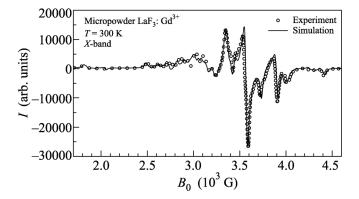
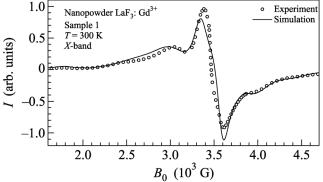


Fig. 3. Electron paramagnetic resonance spectrum of  $Gd^{3+}$  ions in the  $LaF_3$ : $Gd^{3+}$  micropowder.



**Fig. 4.** Electron paramagnetic resonance spectrum of  $Gd^{3+}$  ions in the  $LaF_3$ : $Gd^{3+}$  nanopowder (sample no. 1).

where the isotropic g-factor and crystal field parameters  $B_p^k$  serve as fitting parameters,  $O_p^k$  are Stevens operators [10], and  $\mu_B$  is the Bohr magneton. The coordinate system from [11] was used in calculations: the x axis was oriented along the c axis of the regular LaF $_3$  crystal, and the c axis was chosen along the c axis. Several sets of crystal field parameters close to each other and describing well the electron paramagnetic resonance spectra for the orientations  $\mathbf{B}_0 \parallel x$  and  $\mathbf{B}_0 \parallel z$  are known [11, 12]. It is assumed that the local symmetry of the environment of the impurity  $\mathbf{G}\mathbf{d}^{3+}$  ion substituting for the La $^{3+}$  ion is orthorhombic and there are three types of nonequivalent substitution centers [12]. The latter determines the presence of the additional lines in the spectrum in Fig. 2.

However, as follows from our calculations using the combination of parameters from [11, 12], the agreement with the micropowder spectrum (Fig. 3) containing contributions from all possible orientations of the  $\mathbf{B}_0$  vector with respect to the crystallographic axes is not satisfactory. In order to improve this agreement,

we fitted the parameters  $B_p^k$  and g, which are given in the table. In our calculations, we proceeded from the parameters obtained in [13], where the spectra for the  $\mathbf{B}_0$  orientations in the (xy) and (yz) planes were also considered. As in that work, it occurred that the inclusion of the parameters  $B_n^k$  in the calculation does not improve the agreement with the measured spectra. Therefore, they were taken to be zero. It is remarkable that the parameter  $B_4^{-2} = 0.33 \times 10^{-4} \,\mathrm{cm}^{-1}$  characterizing the local symmetry below orthorhombic is nonzero: attempts at describing the fine details of the micropowder spectrum disregarding this contribution did not succeed. At the same time, the obtained parameter set describes well the spectra of the regular crystal at the orientations  $\mathbf{B}_0 \parallel x$  (Fig. 1) and  $\mathbf{B}_0 \parallel z$ (Fig. 2).

The line shape of the electron paramagnetic resonance spectrum is close to Lorentzian. The question of why it is Lorentzian is a subject of a separate study. For the regular crystal and micropowder, the half-width of all lines was the same and was 21 G (here and below, it

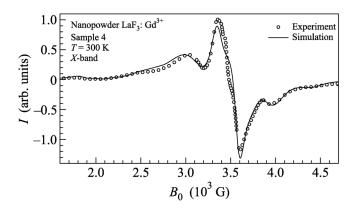
is the half-width of the integral absorption curve). Apparently, the main contribution to the line width in these samples is due to the superhyperfine interaction between the gadolinium ions and the surrounding fluorine ions from the crystal lattice of LaF<sub>3</sub>. Since the sizes of particles in nanopowders are tens of nanometers, the crystal field acting on the impurity center depends on the particle shape and size and the closeness of its boundaries. The spread of the parameters

 $B_p^k$  near their equilibrium values presented in the table occurs and, as a consequence, the spread of the splittings of the energy levels of the ground state of  $\mathrm{Gd}^{3+}$  ions appears. This leads to the increase in the halfwidth of the electron paramagnetic resonance lines, which is different for the central line  $(-1/2 \longleftrightarrow 1/2)$  transition) and all other lines of the spectrum. In fact, it follows from the theoretical considerations in the first order of perturbation theory that the crystal field causes the same shift of the energy of states with the spin projections M and -M on the direction of the field  $\mathbf{B}_0$ :

$$\delta E_{M}^{(1)} = \left\langle M \middle| \sum_{pk} B_{p}^{k} O_{p}^{k} \middle| M \right\rangle$$

$$= \left\langle -M \middle| \sum_{pk} B_{p}^{k} O_{p}^{k} \middle| -M \right\rangle. \tag{2}$$

Consequently, the change of the transition energy  $-1/2 \longleftrightarrow 1/2$  due to the crystal field effect in comparison with its value  $g\mu_B B$  for the free ion is only in the subsequent order of magnitude and its absolute value is much less. The spread of these energies due to the random shifts of the parameters  $B_k^p$  in nanocrystals will be less as well. It is reasonable to introduce two quantities for them,  $\Delta_1$  and  $\Delta_2$ , characterizing the half-widths of the central line and transitions  $\pm 1/2 \longleftrightarrow \pm 3/2$  located close to the central part of the spectrum, respectively; these quantities satisfy the inequality  $\Delta_1 < \Delta_2$ . These two quantities were the only fitting parameters in the calculations of the spectra of nanosized powders (sample nos. 1–4). Within this simple model, no considerable difference between electron paramagnetic resonance spectra of sample no. 1 (in the absence of



**Fig. 5.** Electron paramagnetic resonance spectrum of  $Gd^{3+}$  ions in the  $LaF_3$ : $Gd^{3+}$  nanopowder (sample no. 4).

hydrothermal treatment, Fig. 4), no. 2, and no. 3 was detected, where the half-widths  $\Delta_1$  and  $\Delta_2$  are 5–10 times larger than analogous values for the single crystal. In the spectrum of sample no. 4 with the maximum  $\tau$  value, additional details (Fig. 5) appear because of a some decrease in  $\Delta_1$  and  $\Delta_2$  (to 4–7 half-widths in the single crystal). Obviously, the parameters  $\Delta_1$  and  $\Delta_2$  directly depend on the nanoparticle sizes. The crystal field parameters within the point charge model are given by the sum [14]

$$B_k^p = A_p^k \langle r^p \rangle K_p \gamma_p, \tag{3}$$

where  $\langle r^p \rangle$  is the average value of the *p*th power of the radius of the electron orbit in the ground state of the Gd<sup>3+</sup> ion,  $K_p$  is the reduced matrix element, and  $\gamma_p$  is the screening factor. The parameters  $A_p^k$  are specified by the following sum over the coordinates  $r_i$  of all ligands with charges  $q_i$  in the crystal lattice of LaF<sub>3</sub>:

$$A_{p}^{k} = \sum_{i} q_{i} \frac{Z_{p}^{k}(r_{i})}{r_{i}^{p+1}},$$
 (4)

where  $Z_p^k$  are spherical harmonics [15]. For the regular crystal, the summation over *i* can be expanded to the infinite crystal lattice. For the nanocrystal with the

Crystal field parameters  $B_p^k$  (in  $10^{-4}$  cm<sup>-1</sup>) and the isotropic g-factor of the effective spin Hamiltonian of the ground state of the Gd<sup>3+</sup> ion in the LaF<sub>3</sub> crystal, and the half-width of the electron paramagnetic resonance lines  $\Delta_i^*$  in different samples

Sample	$B_2^0$	$B_2^2$	$B_4^0$	$B_4^2$	$B_4^4$	$B_4^{-2}$	g	$\Delta_1$ , G	$\Delta_2$ , G
Single crystal, micropowder								21	21
Nanopowder nos. $1-3$	79	9.5	0.083	-0.41	0.70	0.33	1.99	100	200
Nanopowder no. 4								80	140

<sup>\*</sup> Indices 1 and 2 refer to the central line (transitions  $-1/2 \leftrightarrow 1/2$ ) and transitions  $\pm 1/2 \leftrightarrow \pm 3/2$ , respectively.

cross section R, this summation is limited by a certain value  $r_i$ . For this  $Gd^{3+}$  ion located in a certain position in the nanocrystal,

$$A_n^{k(\text{nano})} = A_n^{k(\text{mono})} - \delta A_n^k. \tag{5}$$

The shift  $\delta A_p^k$  from the value for the single crystal is due to the contributions from the ligands located at the distances on the order of R and larger. Above, we used the point charge model, which describes adequately the contribution to the crystal field energy from the ligands at a distance exceeding the size of several unit cells. Below, we will limit ourselves to qualitative considerations [16]. Averaging over different positions of the impurity center in the nanoparticle gives  $\langle \delta A_n^k \rangle = 0$ . Consequently, the electron paramagnetic resonance spectra of nanopowders can be described by the same set of parameters  $B_p^k$  as the single-crystal spectra. We ignore the deformation of the nanocrystal structure and the corresponding change in the crystal field parameters as a higher order effect for particle sizes  $R \ge 20$  nm. The spectral line broadening is associated with the presence of the spread of the  $\delta A_p^k$  parameter. The largest contribution to the crystal field comes from the parameter  $B_2^0$  (see table). The half-width owing to the spread of  $B_2^0$  is inversely proportional to R [16]. The ratio of the nanoparticle sizes in sample nos. 1 and 4 is estimated in terms of the half-widths  $\Delta_1$ and  $\Delta_2$  (after subtraction of their single-crystal values) as  $R_1/R_4 = 1.3-1.5$ . This ratio is very close to the corresponding value in the isostructural PrF<sub>3</sub> crystal obtained using electron microscopy [6] (39 nm/27 nm = 1.44).

Our study demonstrates reliably that the method of hydrothermal treatment can be used successfully for the preparation of nanocrystal powders with the particle size varied in a narrow range. The sizes and regularity of the structure of such particles can be determined both directly (using electron microscopy methods) and indirectly (using the estimates of the crystal field parameters and their spread). Good agreement between the calculations and measured electron paramagnetic resonance spectra of Gd<sup>3+</sup> ions in LaF<sub>3</sub>:Gd<sup>3+</sup> indicate the possibility of using the LaF<sub>3</sub>:Gd<sup>3+</sup> nanopowder as a contrast agent in magnetic resonance tomography.

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